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## STABILIZATION OF INDIUM-RICH IN1-XGAXN HETEROSTRUCTURES - THE EXPLORATION OF A C

Nikolaus Dietz  
GEORGIA STATE UNIVERSITY RESEARCH FOUNDATION INC.

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Final Report

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***Stabilization of indium-rich InGaN heterostructures: the exploration of a common processing window***

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***Abstract***

During the grant period, the growth and optimization of indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers grown by high-pressure metalorganic chemical vapor deposition (MOCVD) was explored at reactor pressures from 5 to 20 bar and at growth temperatures of 700-900°C. The goal was to evaluate the reactor pressure and growth temperature relation at which indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers can be stabilized. The results showed that for pressures around 15 bar, the growth temperatures for InGaN varies from 850°C (InN) to 950°C ( $\text{In}_{0.7}\text{Ga}_{0.3}\text{N}$ ), significantly reducing the temperature gap in the ternary InGaN system compared to low-pressure MOCVD. An unexpected side effect found was the significant reduction in growth rate with increasing reactor pressures, which is due to smaller surface diffusion layers at higher pressures. The results on forming single phase indium-rich ternary InGaN alloys using simultaneous and sequential group-III precursor injection sequences was only partial successful: We obtained single phase alloys for  $\text{In}_{1-x}\text{Ga}_x\text{N}$  [ $0 < x < 0.15$ ] and [ $0.25 < x < 0.3$ ] but observed mixed phases for compositions between for the digital alloying concept explored. The experiments indicate the presence of Ga- and/or In-adlayers - and oscillations between them – that may play a major role for the observed mixed InGaN phases. Additional studies will be needed to obtain a better understanding on how the deployment of precursors to the surface relates to the surface decomposition and chemistry processes that influence the Ga- and In-fragment incorporation and the subsequent InGaN phase formation.

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### **Accomplishments - Summary**

During the reporting period, we explored the growth parameter space for indium-rich In<sub>1-x</sub>Ga<sub>x</sub>N layers growth under sequential metalorganic precursor injection and compared the results with simultaneous metalorganic precursor injection, performed in the previous years. The reactor pressure was kept between 8 and 15 bar, with growth temperatures ranging from 700°C - 900°C, depending on the targeted gallium content, which was varied from  $x=0.0$  to  $x=0.4$ . The epitaxial growth was carried out on Sapphire substrates, GaN/Sapphire- and InN/GaN/Sapphire templates.

The growth and analysis of the In<sub>1-x</sub>Ga<sub>x</sub>N epilayers showed that the growth temperature for InN can be increased by 120°C as the reactor pressure increases from 1 to 18 bar. For In<sub>0.85</sub>Ga<sub>0.15</sub>N the growth temperature increases by 110°C in the same pressure range. At the same time however, the growth rate linearly decreases with a slope of 5.6 nm/h for each bar of reactor pressure increase. Significant progress has been made in the growth of InN epilayers grown with various nucleation conditions in order to improve the InN/GaN/Sapphire templates used for InGa<sub>x</sub>N growth. The structural quality of the InN templates varied from 200 to 150 arcsec for the full-width-half-maximum (FWHM) values of the InN(0002) Bragg reflex. The photoluminescence (PL) from these epilayers showed emission peak maxima varying from 0.7eV to 0.95 eV depending on the free carrier concentration analyzed to Fourier transform infrared (FTIR) spectroscopy. The limiting factor in reducing the free carrier concentration in these layers is related to residual oxygen-contamination during the sample loading in at the present reactor, even after an improvised nitrogen glove box attached to the reactor during the loading process was used.

Ternary In<sub>1-x</sub>Ga<sub>x</sub>N layer growth studies to achieve single-phase alloys showed that the pulse separation between the ammonia and the group III-precursor pulse has to be increased with increasing TMG concentration. These studies suggest that the surface reaction chemistry for the group III-precursors - TMI and TMG - differs significantly and contribute to the observed phase instabilities. We explored therefore the growth of ternary In<sub>1-x</sub>Ga<sub>x</sub>N alloys with simultaneous and sequential Indium and Gallium MO injections. A significant effort was dedicated to formulate and rewrite the reactor control software, allowing the sequential indium and gallium precursor injection - denoted as digital alloy injection approach.

The analysis of the In<sub>1-x</sub>Ga<sub>x</sub>N layers grown with this approach showed improved structural quality with no phase segregation observed below 15% gallium incorporation. However, InGa<sub>x</sub>N layers with gallium incorporation higher than 15% showed small second phases, indicating that the parameter space explored so far was not sufficient to eliminate the gallium and/or indium adlayer formation at the growth interface, responsible for the InGa<sub>x</sub>N phase segregation in the grown bulk layers. This can be addressed by expanding the parameter space, evaluating a larger process parameter window that includes pulse separation times, precursor fluxes and flux ratio,

or reactor nitrogen pressures in order to stabilize the difference partial pressures above indium-rich and gallium-rich growth surfaces.

As shown in this final report, substantial progress has been made in the structural quality of indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  epilayers. The HPCVD system employed narrows the presently encounters growth temperature gap by suppressing the decomposition of indium-rich alloys at growth temperatures required for wider band-gap group III-nitrides. Employed real-time optical diagnostics such as Principal-angle-reflectance (PAR) spectroscopy have been utilized to analyze the surface chemistry during nucleation and steady state growth at a sub-monolayer level. Laser light scattering (LLS) was applied to characterize the surface morphology during nucleation and growth. These techniques provided critical insights into gas phase and surface chemistry processes during the HPCVD growth process and helped to adjust the growth process parameters. Further studies are needed to improve the structural, optical and electrical properties of ternary  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers. At this point, we demonstrated that the HPCVD approach allows the stabilization of highly volatile constituents/alloys such as encountered in the growth of indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  epilayers under process conditions not possible by MBE or low-pressure MOCVD.

### Highlights:

- During the research program, three graduate students completed and received their PhD degree and four graduate students completed their MS degree (see section III.2). Three students are still working on various aspects related to this research.
- The research results have been published in 19 referred publications and have been presented in 10 invited publications, 35 oral conference contributions, and 18 conference poster presentations (see section III.3).
- A full national and international patent application, entitled "High pressure chemical vapor deposition apparatuses, methods, and compositions produced therewith," has been filed Aug. 12, 2010 (after provisional filing in Aug. 2010). This patent application describes critical design aspects of a next generation of HPCVD reactor, which integrates discoveries related to research supported during this research program.
- We showed that HPCVD enables the successful growth of high crystalline quality layers of InN on sapphire and GaN/sapphire templates. At a reactor pressure of 15 bar, the growth temperature for InN can be raised to about 850°C. Detailed studies were carried out on the precursor pulse separations and correlated to the crystalline quality of the epilayer. We also carried out extended studies on the optimum group V/III precursor ratio and its influence and the surface chemistry, crystalline quality and the reduction of the free carrier concentration in the layers. A large difference in free carrier concentrations in layers grown on GaN/sapphire templates compared to InN layers grown directly on sapphire was observed.
- The growth results on indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys showed that macroscopic single phase InGaN epilayers can be stabilized. However, significant broadening of the FWHM values in the XRD Bragg reflexes indicate high point defect densities. At present the process conditions have to be adjusted of each composition region in order to stay single phase, requiring more detailed microscopic structural and optical defect studies on potential compositional fluctuations.
- In the compositional region  $0.3 < x < 0.4$  of  $\text{In}_{1-x}\text{Ga}_x\text{N}$ , a significant reduction of the measured free carrier concentration is observed. The reason is not clear and further experimental studies are needed.

## ***I. High-pressure chemical vapor deposition (HPCVD) growth of indium-rich InGaN epilayers***

However, all presently employed low-pressure deposition techniques encounter significant temperature gaps in the growth of binary group III-nitrides. For instance, the optimum growth temperatures of InN and GaN differ by more than 300°C under low-pressure organometallic chemical vapor deposition growth conditions. Such a temperature gap severely limits the ternary InGaN alloy formations and their integration in wider band-gap alloys that have to be grown at higher growth temperatures. One consequence of this problem is discussed in the context of spinodal decomposition/compositional fluctuations in the ternary InGaN<sup>1-8</sup> system – an added problem to compositional induced lattice strain, interfacial piezoelectric polarization effects, and extended defect related effects that have to be addressed.

A potential pathway to address and overcome the difficulties associated with the phase stability, stoichiometry fluctuations and the growth temperature gap between the group III-N binaries, is to assess the pressure dependency of surface chemical reactions and growth surface stabilization, one potential pathway explored at GSU.

### ***High-pressure chemical vapor deposition (HPCVD): Motivation and History***

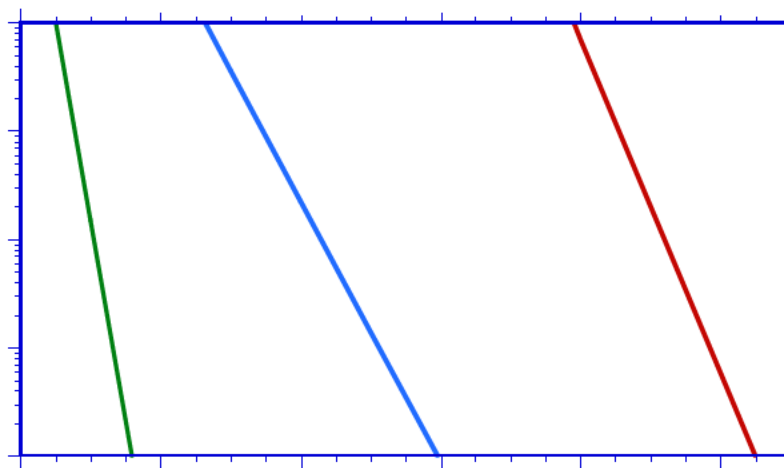
Research on extending OMCVD to super-atmospheric pressure is motivated by the sensitive relation between the properties of compounds and their native defect chemistry. In turn, the defects depend on the control of compound stoichiometry, that is, on the partial pressure of volatile constituents in thermal equilibrium. For many materials utilized in today's industry, the decomposition pressures are sufficiently low to permit processing at reduced pressure, which avoids fluid dynamics perturbations of process uniformity. However, there are important merging materials systems, where stoichiometry control is limited under conditions of low total pressure. For example, limitations are encountered at present in the control of the stoichiometry and defect formation for InN and indium-rich group III-nitride solid solutions in processing at reduced pressure, due to the high decomposition pressure and their vastly different partial pressures. MacChesney et al.<sup>9</sup> assessed within the thermodynamic limitations the growth of high-quality InN, suggesting that high pressures are needed to stabilize the compound. The calculation indicated that substantial nitrogen pressure is required to prevent thermal decomposition of bulk InN, a relationship captured by

$$p(N_2) \rightarrow p_0 \exp \left[ -\frac{\Delta H_f}{R} \left( \frac{1}{T} - \frac{1}{T_0} \right) \right], \quad (1)$$

which results in the  $p$ - $T^{-1}$  relation shown in Fig. 1<sup>9</sup>. This relation suggests that, for pressures  $p_{N_2} \leq 10^2$  bar and substrate temperatures  $\leq 900$  K, the surface decomposition of InN can be effectively suppressed. Also, recent studies in the indium-gallium-nitrogen system<sup>10</sup> show much uncertainty in the  $p$ - $T$ - $x$  relations (where  $x$  stands for Ga/In ratios) due to missing experimental validation.

Even though the transition from bulk crystal growth techniques towards thin film growth techniques (e.g., MBE, MOCVD, MOVPE, CVD, etc. ) opens unique off-equilibrium approaches to stabilize growth surfaces at temperatures and pressures not possible otherwise, the integration of such highly dissimilar alloys remain a main challenge due to miss-matched processing windows or stoichiometric instabilities and low dissociation temperatures that may lead to inconsistent and process dependent material properties.

Keeping this in mind, Dr. Bachmann and Dr. Banks at North Carolina State University (NCSSU) addressed this problem in 1995, in a MURI research program entitled "Modeling and Control of Chemical Vapor Depositions Processes: The Control of Defects in Mixed III-V Compound Heterostructures," and started the modeling and design of reactor systems, suitable to operate at elevated pressures,<sup>11-15</sup>, an effort which was funded by AFOSR under DOD-MURI F49620-95-1-0447.



**Fig. 1:**

Thermal decomposition pressure vs. reciprocal temperature for AlN, GaN and InN<sup>9</sup>.

The research program simulated and analyzed various reactor geometries and provided a theoretical assessment of a well-suited high-pressure CVD flow geometry. Based on the predictions, a flow channel reactor design was singled out. The experimentally constructed differential HPCVD reactor system is depicted in Fig. 2. In order to confine pressures up to 100 bar, a large outer pressure confinement vessel was required, which made its operation very cumbersome and difficult to control. However, over the three years of operation, significant experience was gained in assessing the flow kinetics of the flow channel and the pressure balancing requirements during inserting of precursor plugs in the gas carrier stream. The knowledge accumulated during this time led to the design of a 2<sup>nd</sup>-generation reactor, the construction of which Drs. Bachmann and Dietz started in 1998. The PI completed the reactor at GSU 2001. The involvement of the PI in the MURI project focused initially on the development of real-time process monitoring<sup>16-18</sup> and control methodology<sup>18,19</sup> using Ga<sub>1-x</sub>In<sub>x</sub>P as an example<sup>20-23</sup>. The involvement expanded as the PI closely interacted with Dr. Bachmann in the design and construction of a compact HPCVD reactor, which is schematically shown in Fig. 3. Besides of the drastic reactor size reduction, the most significant advances implemented in the 2<sup>nd</sup>-generation of HPCVD system were

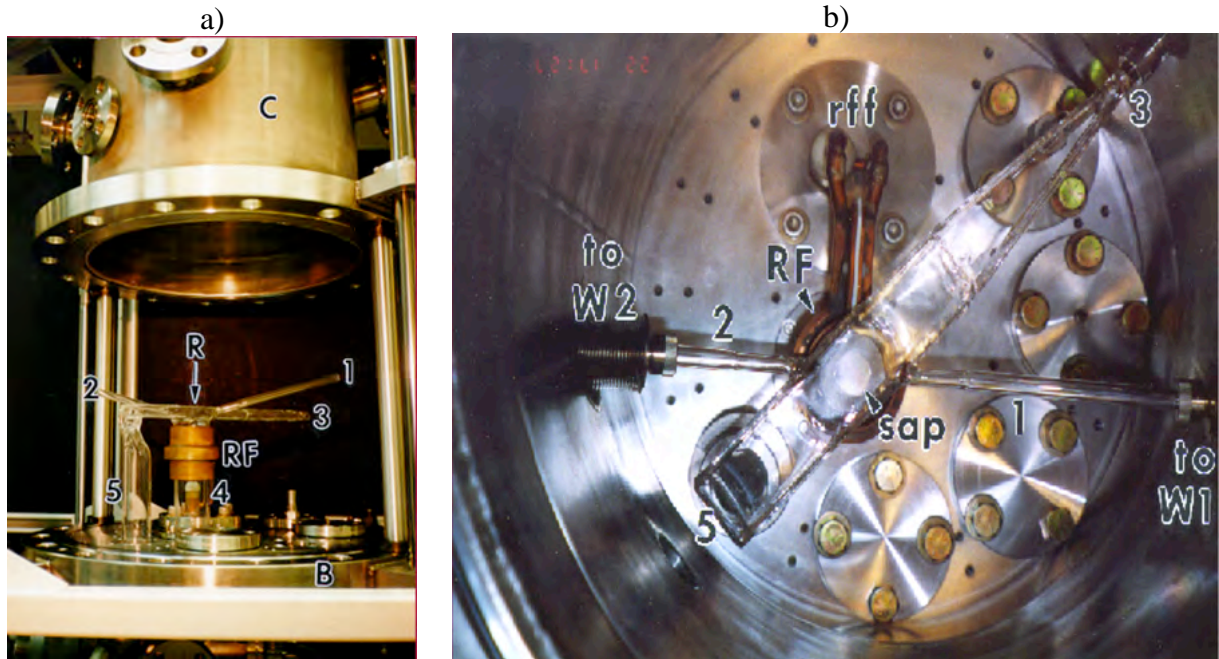
- a) a reduction of the flow channel height from 10 mm to 1 mm,
- b) a symmetric flow channel and substrate arrangement, and
- c) the integration of optical diagnostics for gas phase and growth surface analysis.

A more detailed description of the reactor design and the optical characterization capabilities can be found at ["http://www.phy-astr.gsu.edu/dietzrg/HPCVD.html"](http://www.phy-astr.gsu.edu/dietzrg/HPCVD.html).

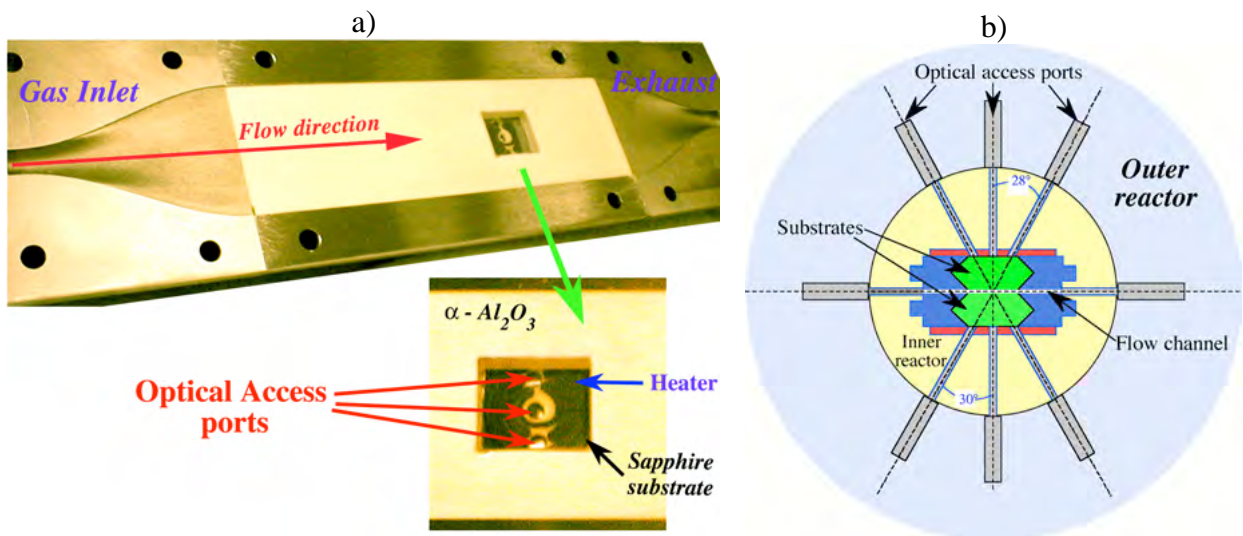
The construction of the compact HPCVD reactor was completed at GSU with support of NASA Grant# NAG8-1686 (from 2000 to 2006; Dr. Bachmann was Co-PI on the project and he retired



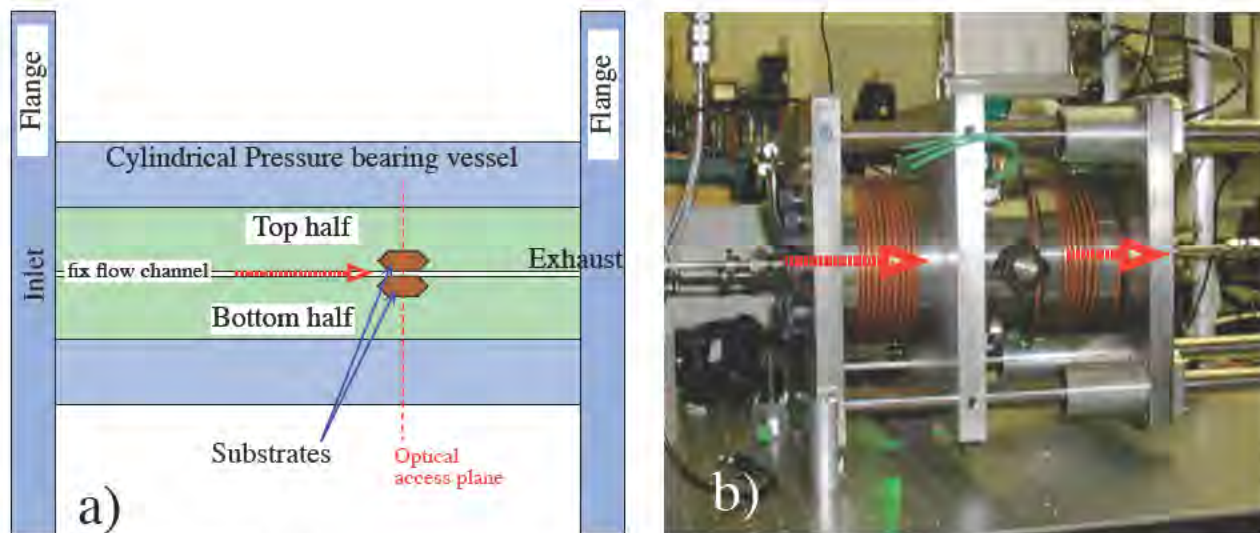
from NCSU in 2003) with a main emphasis on demonstrating the flow kinetics and abilities of the real-time growth diagnostics<sup>24-27</sup>.



**Fig. 2:** a) 1<sup>st</sup> generation HPCVD reactor assembly constructed at NCSU in 1996. B = Base Plate; C = 2nd Confinement Shell; R = Fused Silica Reactor; 1&2 = Window Connections for PRS Laser Beams; RF = Radio Frequency Coil; 3&5 = Process Gas Inlet & Outlet; 4 = Tube on R For substrate wafer exchange. b) top view of inner flow channel assembly.



**Fig. 3** a) 2<sup>nd</sup> generation HPCVD reactor assembly. The flow channel is designed with a constant cross sectional area for the maintenance of laminar flow and the substrates are embedded in ceramic plates; b) Schematic cross section of the reactor containing the optical access ports and the center of the substrates. Optical ports provide access to the flow channel and to the growth surface.



**Fig. 4.** a) Schematic of 2<sup>nd</sup> HPCVD reactor with outer cylindrical pressure confining vessel shown. b) Picture of physical HPCVD reactor built in 2001.

Accessing the growth regime at super-atmospheric pressures brings significant challenges in suppressing gas phase reactions, while controlling the nutrient support through a reduced diffusion layer to the growth surface and optimizing the growth surface chemistry. An essential component in the exploration of high-pressure CVD growth is the integration of real-time optical characterization techniques that allow to monitor and analyze the gas flow kinetics, the precursor decomposition dynamics, as well as growth surface reactions. The PI has a long track record of developing optical diagnostic tools<sup>28,29,23</sup> and of applying them for real-time process monitoring<sup>23</sup> and process control<sup>18,19</sup>. For high-pressure CVD, the PI integrated principal angle reflectance spectroscopy (PARS)<sup>30</sup> - a derivation of p-polarized reflectance spectroscopy (PRS)<sup>16</sup>, which is able to follow the film growth process with sub-monolayer resolution. The link between the surface sensitive PARS response to the gas phase analysis via ultra-violet absorption spectroscopy (UVAS) allows for the link between gas phase decomposition kinetics and surface chemistry processes, which will provide critical insights in the film growth process at high pressures. With support by AFOSR (award# FA9550-07-1-0345), the PI focused in recent years on the optimization of InN and indium-rich InGa<sub>0.9</sub>N growth in the pressure regime of 5 to 15 bars, the results of which are presented in more details below.

#### ***InN and InGa<sub>0.9</sub>N specific material challenges:***

InN is predicted to have an electron affinity of 5.8 eV, the largest of any known semiconductor<sup>31</sup>. The consequences of the large electron affinity of InN and indium-rich InGa<sub>0.9</sub>N can be considered within the amphoteric defect model (ADM). Within the ADM, the formation energy of native defects depends on the location of the Fermi energy ( $E_F$ ) with respect to a common energy reference, the Fermi stabilization energy ( $E_{FS}$ ). Therefore, native donor formation is predicted to be dominant in InN and indium-rich InGa<sub>0.9</sub>N. The low formation energy of native donor defects in InN and indium-rich InGa<sub>0.9</sub>N creates challenges for producing p-type materials<sup>32</sup>. Degenerate doping may be a solution to achieve p-type InN and indium-rich InGa<sub>0.9</sub>N, due to the high n-type background of undoped InN.

The pulsed precursor injection scheme employed in HPCVD to minimize gas phase reactions bears also significant advantages for the prevention of phase segregation and for the exploration of the surface chemistry during growth. An important research aspect is the study and



understanding of the growth kinetics on a micro-/nanometer scale, in order to develop an optimum pulse timing. Once this understanding is established, it can be applied to the digital alloying of InGaN, which will not only allow the prevention of phase segregation but also the fabrication of III-nitride superlattices<sup>33,34</sup>.

For the fabrication of indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys and embedded heterostructures, the thermal stability of InN and indium-rich alloys at growth temperatures that are compatible with GaN growth conditions, needs to be advanced. Our initial InGaN growth results in the pressure regime up to 15 bars indicate that pressures above 20 bars may be required - a regime that provides some technical challenges and has not been investigated to date. In this pressure regime, the role of turbulent gas flow becomes decisive.

Another critical issue is the type of growth mode: 2-dimensional (2-dim) versus 3-dimensional (3-dim) film growth. In 2-dim growth mode, material is deposited layer-by-layer. On the other hand, 3-dim growth consists of formation of islands and their subsequent coalescence. The latter results in grain boundaries that detrimentally influence the topographical and electrical properties of the deposited epilayer, e.g., carrier mobility and free carrier concentration<sup>35</sup>. Good topographic properties of  $\text{Ga}_{1-x}\text{In}_x\text{N}$  layers (i.e., a smooth surface) are essential for the fabrication of heterostructures. Benchmarks of 3-dim growth are the size, shape, height and density of the islands.

The growth mode during the initial stage of epitaxy (nucleation) is of particular interest, as the quality of the epilayer is governed by the quality of the nucleation layer. Therefore, a good understanding and control of the nucleation and nuclei coalescence is decisive. Moreover, the growth of  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys brings along the issue of phase segregation and spinodal decomposition. The here explored high-pressure CVD approach is promising for enabling new Ga/In ratios (i.e., new  $x$  values) that have not been achieved before. Thus phase homogeneity becomes a very important goal of our work. Finally, new alloys may exhibit new defects related to ordering processes in a microscopic scale or cubic/wurtzite lattice instabilities. A careful analysis and identifications of such defects is required. A major effort in our research will pursue a deep understanding of occurring defects, how they differ as a function of growth pressure, and how they affect the materials properties. A beneficial effect of HPCVD is the potential decrease in the native point defect concentrations with increased growth temperature at higher reactor pressures.

The epitaxial growth of ternary III-V systems is characterized by the segregation of one of the constituent column III element at the growth front and at the interfaces with the binary material. This segregation results in poor composition profiles and poor interfacial width control. Chemical stability can be influenced by several factors, a number of which have been explored in the literature, such as heat of formation, ion size, and interfacial strain. The driving force for this segregation in InGaN can be considered to be a replacement reaction of Ga for the In in the substrate. The heat of formation for GaN is  $-156.8$  kcal/mole whereas that for InN is  $-28.6$  kcal/mole. The ejection of In from an underlying InGaN layer with Ga deposition thus results in a lower free energy for the surface. The transport of In to the surface is mediated by the surface exchange of Ga for In. The lower free energy of the GaN layer accounts for the asymmetry in the In diffusion profile with growth order in compositional modulated structures. This preferential segregation may be limited by migration enhanced epitaxy techniques in MBE and by variation of the III/V ratio in MOCVD and CBE. Segregation is also seen with annealing processes and current injection techniques.

Other aspects of scientific interest arise from the strong polarity of Group III-nitride crystals. A higher concentration of indium in InGaN/GaN quantum wells (QW) results in more strain and

more polarization<sup>36</sup>. The quantum confined Stark effect (QCSE) is caused by spontaneous polarization and by a strain induced piezoelectric field. Increasing the indium composition increases the piezoelectric field<sup>37</sup>. The resulting QCSE will cause a blue shift at high current densities moving further away from the desired wavelength, and at lower current densities, efficiency will be low due to charge separation<sup>37-39</sup>.

The challenges given by the stabilization of indium-rich group III-nitride alloys and embedded heterostructures in wide bandgap group III-nitrides (e.g.,  $\text{In}_{1-x}\text{Ga}_x\text{N}$ ) can be addressed by the PI's successful development of an advanced 2<sup>nd</sup>-generation version of the high-pressure growth reactor capable of operating at pressures of up to 100 bar. The pulsed injection of precursor gases is a prerequisite for high-pressure operation and, at the same time, it facilitates control of the growth process on sub-monolayers, as well the thorough investigation of surface processes during growth.

## ***II. Accomplishments***

### ***Growth of InN and InGaN under high-pressure CVD conditions***

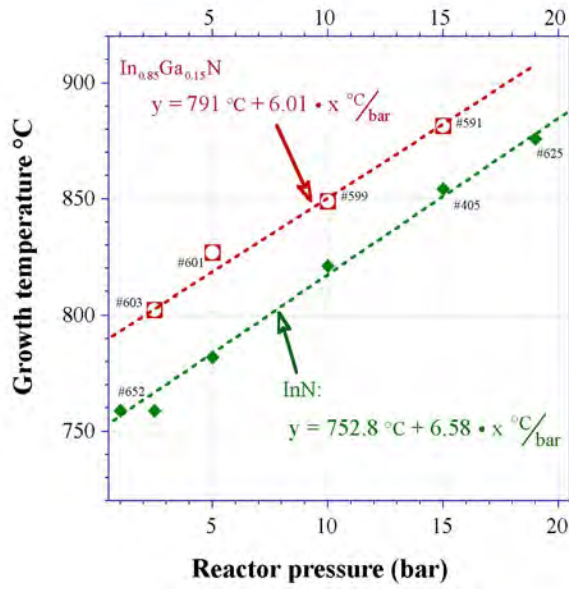
#### **InN layer characterization**

A promising approach to tackle the challenges outlined in the previous section has been developed by the PI at Georgia State University. A unique high-pressure chemical vapor deposition (HPCVD) reactor allows the extension of the thin film growth parameter space by utilizing the pressure dependency<sup>27,40,41</sup> (up to 100 bar) of chemical reactions. Growing indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys at high pressures and high temperatures ( $T > 850^\circ\text{C}$ ) is promising since this approach may overcome problems of off-equilibrium techniques arising from different partial pressures and low growth temperatures. Over the last years our research group demonstrated the capability of HPCVD to produce high-quality, single-phase InN layers. The InN layers exhibit XRD (0002) Bragg reflexes with a full width at half maximum (FWHM) at 150 arcsec and rocking curve values around 1600 arcsec. Further rocking curve analysis for the symmetric and skew-symmetric reflections indicate that - within the experimental resolution - InN grew single phase and epitaxially on the GaN template. A reciprocal XRD map scan shows a nearly relaxed InN epilayer on top of GaN.

#### **InGaN layer characterization**

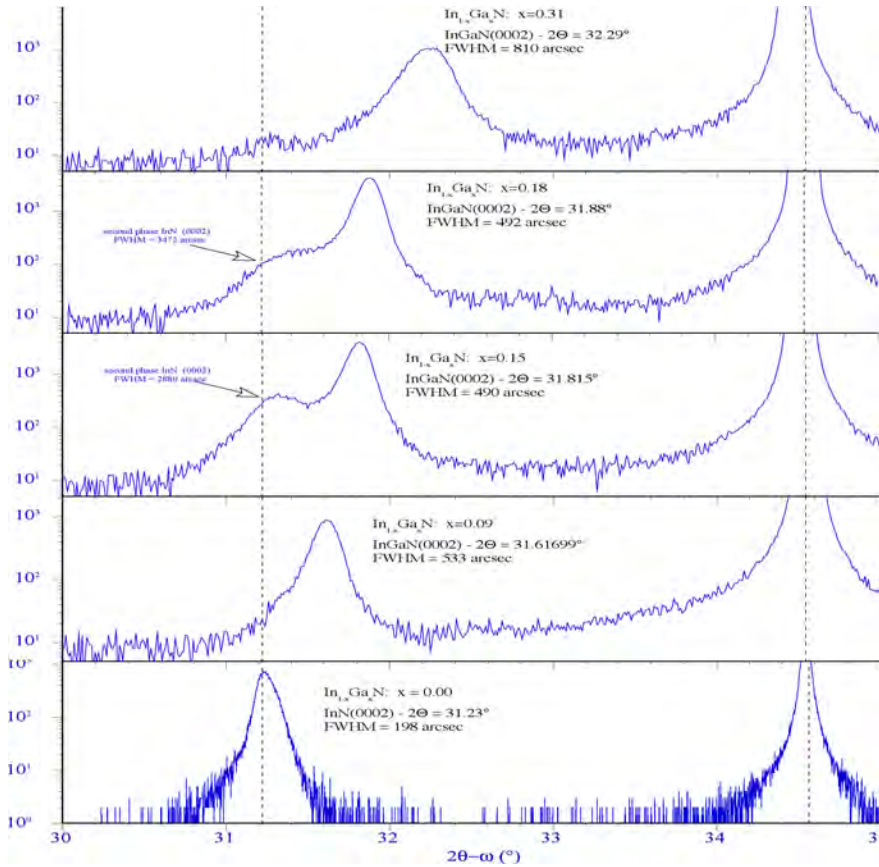
High quality InGaN layers were achieved for growth temperatures in the range of  $830^\circ\text{C}$  to  $850^\circ\text{C}$ , which is about  $250^\circ\text{C}$  higher than under low-pressure OMCVD conditions. As depicted in

Fig. 5, the growth temperatures of InN and InGaN can be increased by more than  $120^\circ\text{C}$  in the pressure regime between 1 and 20 bar, which is a significant advantage over conventional low-pressure metalorganic chemical vapor deposition (MOCVD or OMCVD), where the growth temperatures are around  $650^\circ\text{C}$ . For InGaN, the increase of growth temperature as function of reactor pressure decreases due to the reduced temperature gap between InGaN and GaN. Therefore the MOCVD reactor pressure (high-pressure versus low-pressure) is critical balance between the partial pressures of the alloy compositions that have to be integrated / stabilized at a specific growth temperature.



**Figure 5:** Growth temperature versus reactor pressure for the growth of InGaN epilayers in HPCVD growth conditions (see text). The increased growth temperature narrows the processing gap between the binaries InN and GaN.

The XRD analysis for selected indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers from  $0 < x < 0.65$  are summarized in Fig. 6. Under these pressure and temperature conditions, macroscopic InN-InGaN phase segregations have been observed in the compositional regime  $0.1 < x < 0.30$ , while macroscopic a single-phase material can be obtained in the compositional regime  $0.3 < x < 0.65$ . The  $\omega$ -scan InGaN(0002) rocking curve analysis reveals FWHM's around 5000 – 7000 arcsec ( $x=0.31$ ), indicating a high density of point defects and dislocations.



**Figure 6:**

$\text{In}_{1-x}\text{Ga}_x\text{N}(0002)$  Bragg reflexes of XRD 2Q-w scans for  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layer grown by high-pressure CVD at  $850^{\circ}\text{C}$  and 15 bar reactor pressure.

The  $\text{In}_{1-x}\text{Ga}_x\text{N}$  phase segregations observed differ for  $\text{In}_{1-x}\text{Ga}_x\text{N}$  growth on GaN versus sapphire, indicating that not only the pressure/temperature processing parameter contributes to the

segregation process. Potentially, induced lattice strain, interfacial piezoelectric polarization effects, and extended defects may contribute to the compositional fluctuations.

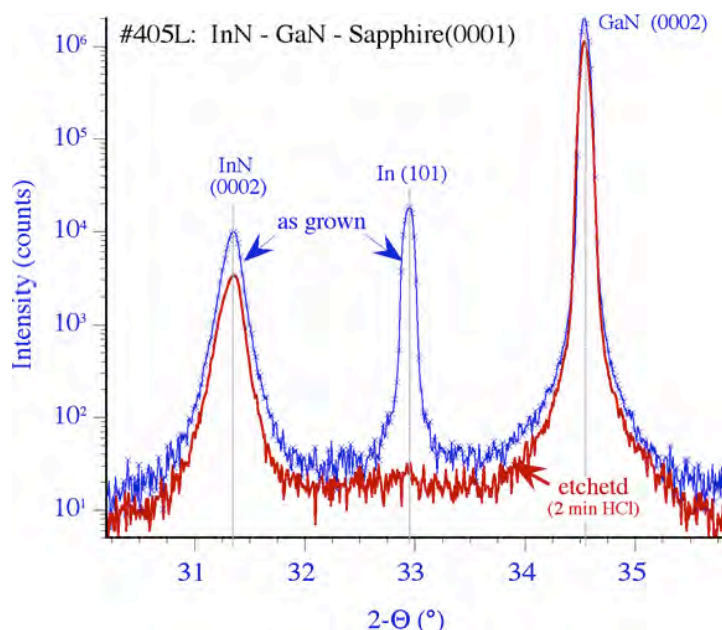
To improve the thermal stability of indium-rich alloys at the desired growth temperatures that are compatible with GaN growth conditions, the growth may have to be expanded to reactor pressures well above 20 bar. Even though this pressure regime inevitably leads to turbulent growth flow conditions, the potential benefits will be the merged temperature processing window that allows the fabrication of indium rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys with wider bandgap group III-nitride layers, an essential step for many of the envisioned device structures.

### Indium add-layer problem

The indium adlayer formation<sup>42-44</sup> during InN and InGaN growth is a well-known phenomenon. Its ability to act as surfactant has been described for the AlGaN/GaN heterostructure growth.<sup>45</sup>

Figure 7 shows the  $2\theta$ - $\omega$  XRD scans for a InN layer grown on a GaN template before and after etching in a HCl:H<sub>2</sub>O (1:10) solution. The In(101) Bragg reflex disappears after typically 2 min etch time, indicated the complete removal of the surface indium.

For the growth of InGaN and InGaN/GaN heterostructures, however, it has to be avoided, requiring precise adjustments of the surface chemistry (precursor pulse separation, growth temperature, reactor pressure). Initial studies during InN growth showed that the indium adlayer formation can be suppressed by adjusting the precursor injection sequence. Detailed studies are required to optimize the surface chemistry for each InGaN target composition.



**Figure 7:**

XRD Bragg reflex of InN(101) is related to an indium adlayer formed during InN growth on a GaN template. The indium add-layer is completely removed by a 2 min HCl:H<sub>2</sub>O (1:10) etch.

### 2.3.2 Real-time growth control and optical growth monitoring

The progress in understanding and controlling thin film growth processes has been very slow, considering how little is known about chemical reaction pathways and reaction kinetics parameters during the decomposition process of the metal-organic (MO) precursors.

These demands led to the development of advanced surface-sensitive optical diagnostics that can be integrated in CVD reactors<sup>46,47,23,26</sup>. These diagnostic techniques move the monitoring and control point close to where the growth occurs which, in a chemical beam epitaxy process, is the surface reaction layer, built up of physisorbed and chemisorbed precursor fragments between the ambient and film interface. In recent years, we developed and explored p-polarized reflectance

spectroscopy (PRS)<sup>23,16,48</sup> as a highly surface sensitive sensing technique, and demonstrated the closed-loop control of deposition processes at low pressure pulsed chemical beam epitaxy<sup>22</sup>. With advancing progress in the growth of indium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$ , the employed optical real-time monitoring techniques will allow for the investigation of fundamental questions regarding surface chemistry. In this context, the competing incorporation of In and Ga atoms is of particular interest for an understanding of compositional questions and segregation processes. During the growth of  $\text{In}_{1-x}\text{Ga}_x\text{N}/\text{GaN}$  heterostructures, we will be able to investigate the physical and chemical processes during the transition from indium-rich to gallium-rich  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers which govern the quality of such heterostructures and which will bring upon clarity about the interfacial phenomena discussed above.

The HPCVD reactor utilizes real-time optical diagnostic techniques - as well as a pulsed precursor injection scheme - to gain insights and to control the gas phase and surface chemistry processes that govern the growth of indium-rich group III-nitride alloys. This is of crucial importance for understanding and controlling their materials properties. We deployed “principal angle reflectance spectroscopy” (PARS),<sup>30</sup> and ultra-violet absorption (UVA) spectroscopy to analyze the kinetics of gas phase constituents above the growth surface.<sup>26</sup> The link between the surface sensitive PARS response to the real-time gas phase analysis (UVAS) provided insights in the gas phase decomposition kinetics, surface chemistry processes, and the film growth process at high pressures. Advanced growth models as established for the growth of  $\text{GaInP}$ <sup>23</sup> is essential for the exploration of high-pressure growth process parameters. Adjusting the pulse separations between the precursors - as well as the length of each precursor pulse - are additional process control parameters that can be utilized in optimizing surface chemistry and materials properties.<sup>49</sup>

### **InGaN gas phase and surface chemistry at elevated reactor pressures**

The formation of  $\text{In}_{1-x}\text{Ga}_x\text{N}$  ternary alloys in the whole composition range is of great interest, since it would allow to tune the direct bandgap from the near infrared (InN around 0.7 eV) to the near UV wavelength regions (GaN at 3.5 eV). However, experimental and theoretical predictions indicate that the  $\text{In}_{1-x}\text{Ga}_x\text{N}$  ternary alloys might be unstable with a tendency toward clustering and phase separations.<sup>50</sup> For instance, it is well known that indium phase separation (or fluctuation) induced localized states in the InGaN layers play major roles in achieving highly efficient blue and green InGaN multiple quantum wells (MQW).

The large differences in the tetrahedral radii between InN and GaN may induce strain that can either lead to the formation of particular sublattices (phase separations) or to an atomic ordering within the sublattice, resulting in a deviation from homogeneity (nano-clustering).<sup>51,50</sup>

Nevertheless, the growth of single phase  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys by rf-PMBE at growth temperatures between 400-435°C has been demonstrated by Iliopoulos et al.<sup>52</sup> in the entire composition range, which suggests that under proper processing conditions, clustering and phase separations in the ternary InGaN alloy system can be suppressed. Under low-pressure MOCVD growth conditions with typical growth temperatures between 700 and 800 °C, metastable  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys are predicted for regions of low and high gallium concentrations ( $0.94 > x < 0.64$  and  $0.1 > x < 0.3$ ) and compositional unstable  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloy regions, where phase separations occur due to spinoidal decomposition.<sup>50</sup> Contrary to these predictions, recent  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers grown by MOCVD indicate that single phase  $\text{In}_{1-x}\text{Ga}_x\text{N}$  alloys in the compositional range  $0.33 > x < 0.75$  can be achieved by adjusting the growth temperature as function of composition.<sup>53,54</sup>

The question left open is to which extend a processing window exists where  $\text{In}_{1-x}\text{Ga}_x\text{N}$  layers with different compositions can be stabilized at the same growth temperature. The high-pressure

CVD reactor system - together with the digital injection system – explored in this research indicated a potential pathway to establish such common processing window, by

- *adjusting the reactor pressure to stabilize a compositional alloy at the temperatures at which the alloy would either decompose or exhibit phase separation, and by*
- *adjusting the group V/III precursor ratio and surface chemistry as function of composition  $x$  with sub-monolayer precision.*

The present 2<sup>nd</sup> generation of HPCVD reactor will have to be modified in to integrate a research results over the last 15 years and to address the main limitations in the present HPCVD system, which are

- a) *lack of a load lock system,*
- b) *1 sq-inch substrate size restriction,*
- c) *precursor intermixing in the flow channel due to run-time issues, and*
- d) *fixed flow channel height.*

Even though the cylindrical pressure vessel design employed in the 2<sup>nd</sup> HPCVD reactor system is ideally suited for reactor pressures well above 100 bars, it became apparent in our research studies that the loading and unloading of the substrates is too complex and exposes the inner reactor to atmosphere after each growth run – even if a glove box enclosure was used. Even if the reactor was purged several times after the substrate loading, the remaining impurity contaminations limit at present the quality of the grown InGaN epilayers.

The next - 3<sup>rd</sup> generation HPCVD design – will have to integrate reactor design elements formulated in our patent to address the above mentioned shortcomings. It should be able to handle ø2-inch substrates, which are loaded via a load-lock station. The outer pressure vessel of formed by a Stainless Steel (SS) block with a rectangular inner cutout, in which the inner MOCVD reactor will be embedded. All components of the pressures confining outer reactor consists of machined and bolted together components, avoiding any welded parts. The outer reactor has to be designed to maintain reactor pressures of up to 25 bars. Both – the lower part of the pressure vessel as well as the top flange – are water cooled to maintain well-defined process conditions. The loading and unloading of the samples is done via a load-lock station, such that the outer pressure vessel reactor and the embedded inner MOCVD reactor of the HPCVD system isn't exposed to atmosphere during the sample transfer process. This will drastically reduce the impurity level in the HPCVD system and will lead to improved InGaN epilayer structures.

The most critical modification in the 3<sup>rd</sup> generation HPCVD reactor addresses the MO precursor injection and gas flow kinetics to engineer the gas- and surface chemistry during ternary III-N alloy formation. As schematically depicted in Fig. 8, the adjustable center flow channel is formed through a lower base plate and an upper u-shaped element clung around the lower base (the upper cutout for the showerhead injection elements is not shown in Figure). This allows for an adjustable height of the flow channel as function of reactor pressure. The showerhead element(s) are embedded on the upper u-shaped flow channel as schematically illustrated in Fig. 9(a) and (b). The additional vertical injection element contains engineered injection ports/areas above and/or in front of the reaction zone. These showerhead elements are shaped to obtain/tailor a desired precursor concentration profile in the reaction zone. Since the upper flow channel position is adjustable, a flexible SS-bellow decouples the push flow from the MO-precursor injection element.



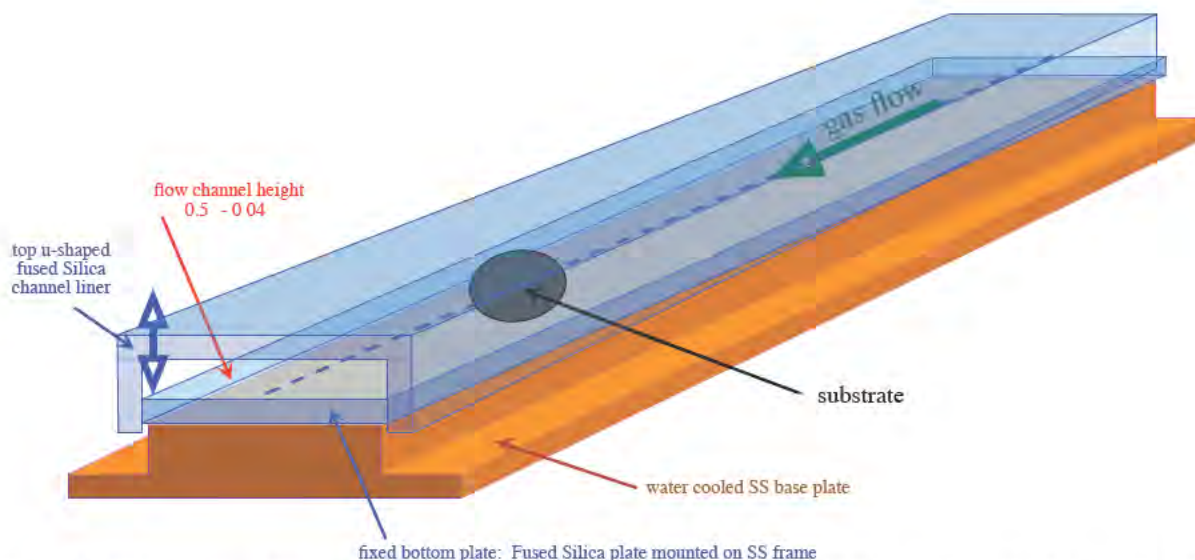
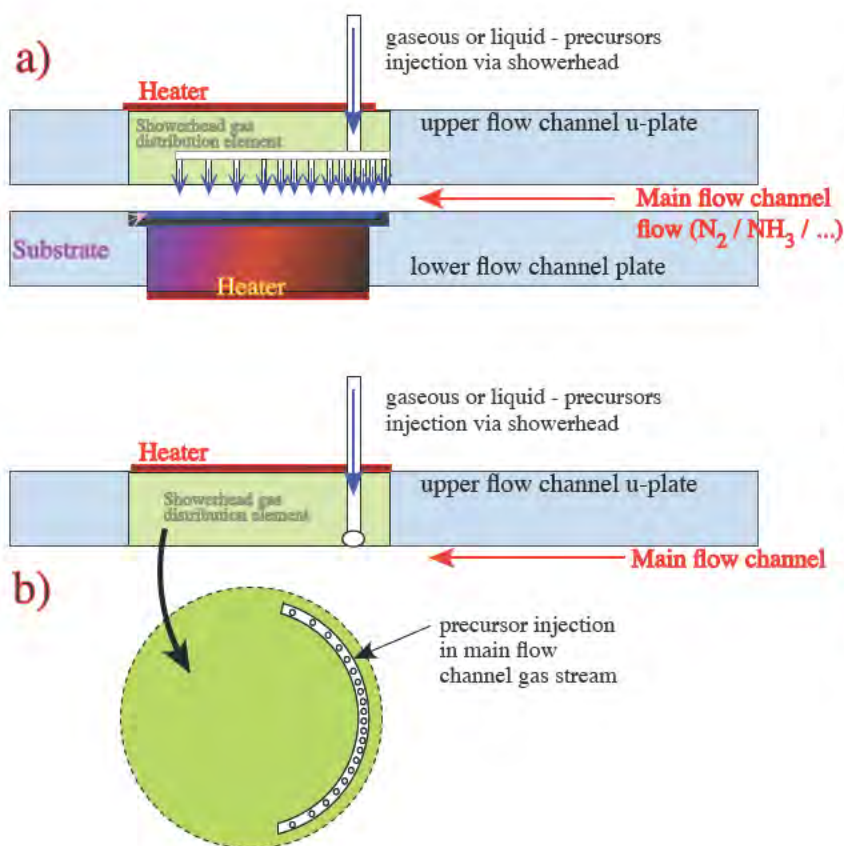


Figure 8. Schematics of the center section of the inner 3<sup>rd</sup>-gen. HPCVD reactor, which consist of an adjustable flow channel height arrangement (from 10 mm down to 1 mm).



**Figure 9.** Schematic illustration of showerhead arrangements for the injection of the MO-precursors perpendicular to the flow channel above and/or in front of the growth surface.

- a) spatial engineered area injection element above reaction zone,
- b) line injection in front of the reaction zone, perpendicular to flow channel.

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<http://dx.doi.org/10.1063/1.3081123>

### ***III. Patent, Theses, Publications and Presentation of research results***

#### ***III.1 Patent filed***

"High pressure chemical vapor deposition apparatuses, methods, and compositions produced therewith," Nikolaus Dietz, (provisional filing Aug. 12, 2009); Application# 61/233,238; WIPO Publication No. WO/2011/019920 Publication date Feb. 17 2011; European Patent # 2,464,760 publ. June 20, 2012.

#### ***III.2.1 Graduate Students supported during award period and still working on...***

**Ms. Indika Senevirathna**, graduate PhD-student: June 2014 - open; "optical characterization of III-N alloys by IR reflectance"

**Mr. Daniel Seidlitz** (MS from Technical University Berlin);

Research PhD-student scholar: Jan 2014 - open; "Optical diagnostics and real-time characterization of MOCVD growth of group III-nitrides"

**Mr. Mark Veron**, MS-student: May 2014 – open; "Growth and analysis of InGaN alloys"

#### ***III.2.2 Graduate Students supported during award period with completed theses:***

**Ms. Jielei Wang** 2008 - Dec. 2011

MS Title: ""Optical Properties of In<sub>1-x</sub>Ga<sub>x</sub>N epilayers grown by HPCVD"  
[http://digitalarchive.gsu.edu/phy\\_astr\\_theses/9](http://digitalarchive.gsu.edu/phy_astr_theses/9)

**Mr. Ramazan Atalay** Aug. 2006 - Nov. 2012

PhD Title: "Optical and Structural Properties of InN epilayers grown by High-Pressure Chemical Vapor Deposition"

**Mr. Max Buegler** March 2008 - Sept. 2012

PhD Title: "Optical and structural properties of Indium-Nitride epilayers and their growth by High Pressure Chemical Vapor Deposition"

**Mr. Ananta Acharya** Aug. 2009 – July 2013

PhD Title: "Indium nitride surface structure, desorption kinetics and thermal stability"  
PhD-Thesis completed Aug. 2013

**Ms. Indika Senevirathna** Aug. 2011 – June 2014

MS Title: "Properties of Group III-Nitride Materials studied using FTIR Reflection Spectroscopy," MS-Thesis completed June 2014; continued PhD program

**Mr. Sampath Gamage** Aug. 2010 – Feb. 2015

MS Title: "Growth and characterization of InN and In-rich InGaN alloys by HPCVD"  
MS-Thesis completed Nov. 2014

**Mr. Rasanga Samaraweera** Aug. 2011 – Feb. 2015

MS Presentation: "Growth and analysis of InGaN alloys and epilayers"  
MS-Thesis completed Jan. 2015

#### ***III.3 Referred Publications (published) during award period:***

1. "Effect of reactor pressure on optical and electrical properties of InN films grown by high-pressure chemical vapor deposition," M. Alevli, N. Gungor, S. Alkis, C. Ozgit-Akgun, I. Donmez A.K. Okyay, S. Gamage, I. Senawirathne, N. Dietz and N. Biyikli, Dietz, Phys. Stat. Sol. C, in print (2015).
2. "A Near-Infrared Range Photodetector Based on Indium Nitride Nanocrystals Obtained Through Laser Ablation," B. Tekcan, S. Alkis, M. Alevli, N. Dietz, B. Ortaç, N. Biyikli, and A.K. Okyay, IEEE Electron Device Lett. **35** (9) pp.936-9, Sept. 2014; doi: 10.1109/LED.2014.2336795

3. "Enhanced memory effect via quantum confinement in 16 nm InN nanoparticles embedded in ZnO charge trapping layer," N. El-Atab, F. Cimen, S. Alkis, B. Ortaç, M. Alevli, N. Dietz, A.K. Okyay, and A. Nayfeh, Appl. Phys. Lett. **104**(25), p.253106 (2014); doi.org/10.1063/1.4885397
4. "The Group III-Nitride Material Class: from Preparation to Perspectives in Photoelectrocatalysis," R. Collazo and N. Dietz, book chapter 8 in 'Photoelectrochemical Water Splitting: Issues and Perspectives,' ed. H-J. Lewerenz and L.M. Peter, RSC Publishing, pp. 193-222, (2013). <http://dx.doi.org/10.1039/9781849737739-00193>
5. "Thermal stability of InN epilayers grown by high pressure chemical vapor deposition," A. Acharya, S. Gamage, M. Senevirathne, M. Alevli, K. Bahadir, A. Melton, I. Ferguson, N. Dietz, and B. Thoms; Appl. Surf. Sci. **268**, p.1 (2013). doi.org/10.1016/j.apsusc.2012.10.184
6. "Room Temperature GaN-based Spin Polarized Emitters," A. G. Melton, B. Kucukgok, Z. Liu, N. Dietz, N. Lu and I. T. Ferguson, Proc. SPIE Vol. **8631**, pp.863104-1-9 (2013). doi: 10.1117/12.2012586
7. "Development of indium-rich InGaN epilayers for integrated tandem solar cells," A. G. Melton, B. Kucukgok, B-Z. Wang, N. Dietz, N. Lu and I. T. Ferguson, Mater. Res. Soc. Symp. Proc. Vol. **1493**, E15.02, pp.1-6 (2013). doi:10.1557/opl.2013.229
8. "Room Temperature GaN-based Spin Polarized Emitters," A. G. Melton, B. Kucukgok, Z. Liu, N. Dietz, N. Lu and I. T. Ferguson, Proc. SPIE Vol. **8631**, pp.863104-1-9 (2013). doi: 10.1117/12.2012586
9. "Development of indium-rich InGaN epilayers for integrated tandem solar cells," A. G. Melton, B. Kucukgok, B-Z. Wang, N. Dietz, N. Lu and I. T. Ferguson, Mater. Res. Soc. Symp. Proc. Vol. **1493**, E15.02, pp.1-6 (2013). doi:10.1557/opl.2013.229
10. Thermal stability of InN epilayers grown by high pressure chemical vapor deposition," A. Acharya, S. Gamage, M. Senevirathne, M. Alevli, K. Bahadir, A. Melton, I. Ferguson, N. Dietz, B. Thoms, Appl. Surf. Sci. **268**, p.1 (2013). doi.org/10.1016/j.apsusc.2012.10.184
11. "Effect of nucleation period on the physical properties of InN epilayers," S. Gamage, M. K. I. Senevirathna, R. Atalay, A. G. U. Perera, A. G. Melton, I. T. Ferguson and N. Dietz, Proc. of SPIE Vol. **8484** pp.84841I-5 (2012). doi.org/10.1117/12.930363
12. "Effect of V/III molar ratio on the structural and optical properties of InN epilayers grown by HPCVD," R. Atalay, M. Buegler, S. Gamage, M. Senevirathna, B. Küçükğök, A. Melton, A. Hoffmann, A. Perera, I. Ferguson and N. Dietz, Proc. of SPIE Vol. 8484 pp.84840X-8 (2012). doi.org/10.1117/12.930199
13. "Effect of reactor pressure on the electrical and structural properties of InN epilayers grown by high-pressure chemical vapor deposition," M. Senevirathna, S. Gamage, R. Atalay, A. R. Acharya, A. Perera, N. Dietz, M. Buegler, A. Hoffmann, L. Su, A. Melton, and I. Ferguson, J. Vac. Sci. Technol. A **30**(3), pp.031511-6 (2012). doi:10.1116/1.4705727
14. "Observation of NH<sub>2</sub> species on tilted InN(01-11) facets," A. R. Acharya, M. Buegler, R. Atalay, N. Dietz, B. D. Thoms, J.S. Tweedie and R. Collazo, J. Vac. Sci. Technol. A **29**(4) pp.041402-5 (2011).
15. "Growth temperature and growth rate dependency on reactor pressure for InN epilayers grown by HPCVD," M. Buegler, S. Gamage, R. Atalay, J. Wang, M. K. I. Senevirathna, R. Kirste, T. Xu, M. Jamil, I. Ferguson, J. Tweedie, R. Collazo, A. Hoffmann, Z. Sitar, and N. Dietz, Phys. Stat. Sol. (c) **8** pp. 2059-2062 (2011).



16. “Reactor pressure - growth temperature relation for InN epilayers grown by high-pressure CVD,” M. Buegler, S. Gamage, R. Atalay, J. Wang, I. Senevirathna, I. Ferguson, R. Collazo, A. Hoffmann, Z. Sitar, and N. Dietz, Proc. of SPIE Vol. 7784, doi: 10.1117/12.860952, 77840F-1-7 (2010).
17. “Optical Properties of InN Grown on Templates with Controlled Surface Polarities,” R. Kirste, M.R. Wagner, A. Strittmatter, J. H. Schulze, R. Collazo, S. Sitar, M. Alevli, N. Dietz and A. Hoffmann, physica status solidi (a), 207(10) pp. 2351–2354 (2010).
18. “The effects of V/III molar ratio on structural properties of In<sub>0.65</sub>Ga<sub>0.35</sub>N layers grown by HPCVD,” G. Durkaya, M. Buegler, R. Atalay, I. Senevirathne, M. Alevli, O. Hitzemann, M. Kaiser, R. Kirste, A. Hoffmann, and N. Dietz, physica status solidi (a) 207, pp. 1379 (2010).
19. “Growth temperature - phase stability relation in In<sub>1-x</sub>Ga<sub>x</sub>N epilayers grown by high-pressure CVD,” G. Durkaya, M. Alevli, M. Buegler, R. Atalay, S. Gamage, M. Kaiser, R. Kirste, A. Hoffmann, M. Jamil, I. Ferguson, N. Dietz, Mater. Res. Soc. Symp. Proc. 1202, paper# 1202-I05-21, pp.1-6 (2010).

#### **III.4 Presentations at conferences/seminars during award period:**

##### **Invited Presentation:**

1. “Development and integration of indium-rich group III-nitrides for energy generation/utilization,” N. Dietz, 11th IEEE International Conference HONET-PfE (Photonics for Energy and Enabling Technologies),” HS2-2, Monday Dec. 15, 2014 @ 13:50, UNC Charlotte; Charlotte, NC; [http://www.honet.uncc.edu/speakers/Nikolaus\\_D.htm](http://www.honet.uncc.edu/speakers/Nikolaus_D.htm)
2. “Perspectives of Group III-Nitride Material for Photoelectrocatalysis,” N. Dietz, 225th ECS Meeting, session “Electronic and Photonic Devices and Systems - Q2: Wide Bandgap Semiconductor Materials and Devices 15,” #Q2-1533, May 14, 2014 @ 8:45am, Hilton Orlando, Orlando, FL (2014); <https://ecs.confex.com/ecs/225/webprogram/programs.html>
3. “The development of indium-rich InGa<sub>N</sub> epilayers and heterostructures,” Nikolaus Dietz seminar lecture at SFB 787 on “Semiconductor - Nanophotonics,” Technical University Berlin, Institute of Solid State Physics; Berlin Germany, July 09<sup>th</sup> (2013).
4. “Group III-Nitride Materials Research for Renewable Energy Use,” Nikolaus Dietz, lecture at workshop entitled “Nanoscience & Nanotechnology for Renewable Energy Applications” at the “International Workshop on Cleanroom training for critical & sustainable technologies: Renewable Energy,” Bilkent University - UNAM, Ankara, Turkey, Thursday, June 27th (2013).
5. “Physical principles of group III-V thin film growth and growth monitoring,” Nikolaus Dietz, invited lecture in the Department of Physics at Marmara University, Istanbul Turkey, June 24th (2013).
6. “Growth and characterization of indium-rich InGa<sub>N</sub> epilayers,” Nikolaus Dietz, Institute seminar Tuesday, Sept 04, 2012, at Helmholtz-Zentrum Berlin (HZB), Institute Solar Fuels; Berlin Germany (2012).
7. “Growth and characterization of indium-rich InGa<sub>N</sub> epilayers grown by high-pressure CVD,” Nikolaus Dietz, Twelfth International Conference on Solid State Lighting; Conference OP220; August 2012, San Diego, CA, contribution# 8484-30, Aug. 16 (2012).

8. “InGaN materials research and applications towards new energy sources,” Nikolaus Dietz, Helmholtz-Zentrum Berlin (HZB), Institute Solar Fuels, Wednesday, July 13, 2011; Berlin Germany (2011).
9. “Materials Research and Applications towards New Light and Energy Sources,” Nikolaus Dietz, invited keynote presentation at International Forum on Advanced Materials and Commercialization, Shangri-La Hotel Ningbo, China, November 9 - 12, 2010.
10. “The pursuit on narrowing the growth temperature gap for InGaN heterostructures,” Nikolaus Dietz, R. Atalay, M. Buegler, S. Gamage, I. Senevirathna, J. Wang, Symposium G at E-MRS Spring Meeting 2010 - Strasbourg, France, June 7-11, 2010.

#### *Conference - oral contributions*

1. “Enhanced Retention Characteristic of MOS Charge Trapping Memory with InN Nanoparticles Embedded in ZnO Charge Trapping Layer,” N. El-Atab, F. Cimen, S. Alkis, B. Ortac, M. Alveli, N. Dietz, A.K. Okyay, A. Nayfeh; MSR Fall Meeting 2014 presentation M1.05 @9:45am, session : AA6: Dislocation Structure and Relaxation,” Dec. 1, 2014, Symposium M: Materials and Technology for Nonvolatile Memories (2014)
2. “MEPA-MOCVD Growth of GaN/GaInN Epilayers and their Structural and Optoelectronic Properties,” D. Seidlitz, R. Kirste, R. Samaraweera, M. R. Bobea, Z. Sitar, N. Dietz, R. Collazo, and A. Hoffmann; presentation AA6.05 @11:45 AM , session : AA6: Dislocation Structure and Relaxation,” Dec. 3, 2014, Symposium AA: Synthesis, Processing and Mechanical Properties of Functional Hexagonal Materials (2014)
3. “Growth Template Impact on the Properties of InN Epilayers Grown by High-Pressure CVD,” N. Dietz, S. Gamage, M.K.I. Senevirathna, R. Kirste, R. Collazo, B. Hussain, I.T. Ferguson; Paper# EM+EN-FrM11, Fri. Nov. 14, 2014 @ 8:20am, session “Nitrides for LED and PV Device Applications,” AVS 61st International Symposium; Baltimore, MD, Nov. 9-14 2014.
4. “Real-time InGaN growth monitoring during plasma-assisted MOCVD,” D. Seidlitz, R.L. Samaraweera, B. Hussain, I. Ferguson, and N. Dietz, Symp. SPIE Opt. Eng. & Appl., Session 7: III-Nitride LEDs for SSL, Aug. 21, 2014 at 2 pm, SPIE Paper# 9190-29, 13<sup>th</sup> Int. Conf. on SSL and LED-based Illumin. Systems; San Diego, CA (2014).
5. “Property analysis of InGaN layers grown by remote-plasma assisted MOCVD,” R.L. Samaraweera, D. Seidlitz, I. M. Senevirathna, B. Kucukgok, B. Hussain, I. Ferguson, and N. Dietz, Symp. SPIE Opt. Eng. & Appl., Session 7: III-Nitride LEDs for SSL, Aug. 21, 2014 at 2:40 pm, SPIE Paper# 9190-31, 13<sup>th</sup> Int. Conf. on SSL and LED-based Illumin. Systems; San Diego, CA (2014).
6. “InGaN growth studies using migration-enhanced, remote-plasma MOCVD,” R. L. Samaraweera, D. Seidlitz, M.K.I. Senevirathna, B. Kucukgok, B. Hussain, I. Ferguson, and N. Dietz, 5th Int. Conf. White LEDs and Solid State Lighting (WLED-5), Presentation, Paper# A1028, June 1- 5, Samdo2-dong, Jeju City, Jeju-do, Korea (2014); <http://www.wled5.org/>
7. “Effect of Reactor Pressure on The Optical And Electrical Properties of HPCVD Grown InN Films,” M. Alevli, C. Ozgit-Akgun, I. Donmez, A. K. Okyay, N. Biyikli, S. Gamage, I. Senevirathna and N. Dietz, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), presentation#G1, contr#1966892, May 20, 12:15-11:45am, Atlanta, GA (2014).



8. [“Advances in migration-enhanced, plasma-assisted MOVCD growth of InGaN epilayers,”](#) Nikolaus Dietz, ‘Workshop on Compound Semiconductor Materials and Devices,’ WOCSEMMAD 2014,’ February 16-19, 2014, San Antonio, Texas.
9. [“The growth and structural properties analysis of indium-rich InGaN epilayers,”](#) S. Gamage, K. Nanayakkara, M.K.I. Senevirathna, A. Melton, I. Ferguson, and N. Dietz, session EM+NS+SS+TF-FrM9 at 11:20am, Nov. 01, 2013; 60<sup>th</sup> AVS Int. Symposium. Long Beach CA (2013).
10. [“InGaN epilayer growth using remote-plasma MOCVD,”](#) R. L. Samaraweera, F. Güth, K. Nanayakkara, M.K.I. Senevirathna and N. Dietz, session EM+NS+SS+TF-FrM9 at 11:00am, Nov. 01, 2013; 60<sup>th</sup> AVS Int. Symposium. Long Beach CA (2013).
11. [“Is GaN:Gd a Viable Route for Spin Polarized Emitters?”](#) A. G. Melton, B. Kucukgok, Z. Liu, N. Dietz, N. Lu and I. T. Ferguson, NSF-Workshop: US-Japan Frontiers in Novel Photonic-Magnetic Devices, Kasugano-so, Nara, Japan, September 20-23, (2013).
12. [“The exploration of InGaN based incoherent type-III heterostructures,”](#) Nikolaus Dietz, I. Ferguson, and R. Tsu, ‘Workshop on Development of Man Made Electronic Materials and Devices: Past and Future,’ May 5-7, 2013; UNC Charlotte, NC
13. [“HPCVD and Migration-enhanced, remote-plasma MOVCD growth of group III-Nitride epilayers,”](#) Nikolaus Dietz, ‘Workshop on Compound Semiconductor Materials and Devices 2013,’ February 17-20, 2013, New Orleans, LA
14. [“Dependence of Gallium Incorporation and Structural Properties of indium-rich  \$\text{In}\_x\text{Ga}\_{1-x}\text{N}\$  Epilayers on Ammonia - MO Precursor Pulse Separation,”](#) S. Gamage, R. Atalay, M.K.I. Senevirathna, R.L. Samaraweera, A. Melton. I. Ferguson, and N. Dietz, Thur. Nov. 01, 2012 - ThA-11 (Electronic Materials and Processing) at AVS 59th International Symposium, Tampa, FL, 2012.
15. [“Effect of V/III molar ratio and reactor pressure on the optical properties InN,”](#) M. K. I. Senevirathna, S. Gamage, R. L. Samaraweera, R. Atalay, A. G. U. Perera, A. Melton, I. Ferguson, and N. Dietz, Thur. Nov. 01, 2012 - contribution ThA-09 (Electronic Materials and Processing) at AVS 59th International Symposium, Tampa, FL, 2012.
16. [“Effect of V/III molar ratio and reactor pressure on the optical properties InN”](#) R. Atalay, M. Buegler, S. Gamage, M. K. I. Senevirathna, B. Kucukgok, A. G. Melton, I. T. Ferguson, and N. Dietz; Twelfth International Conference on Solid State Lighting; Conference OP220; August 2012, San Diego, CA, contribution# 8484-30, Aug. 16 (2012).
17. [“Growth and characterization of indium-rich InGaN epilayers”](#) Nikolaus Dietz, ‘The Workshop on Compound Semiconductor Materials and Devices - WOCSEMMAD 2012,’ February 19-22, 2012, Napa Valley, CA
18. [“The effect of ammonia - TMI pulse separation on the structural properties of InN epilayers,”](#) Ramazan Atalay, M. Buegler, S. Gamage, I. Senevirathna, U. Perera, J. Tweedie, R. Collazo, and N. Dietz, Mon. Oct. 31, 2011 - Paper EM1-MoA-4 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31 - Nov. 04, 2011.
19. [“Digital precursor injection approach for improved indium-rich InGaN epilayers grown by HPCVD,”](#) Nikolaus Dietz, Paper# 8123-10, Monday Aug. 22 at 5:15pm; Conference 8123, Eleventh International Conference on Solid State Lighting, SPIE 2011, 21 - 25 August 2011, San Diego, CA.

20. [“The influence of ammonia precursor exposure and separation times on the structural properties of InN grown by pulsed,”](#) Max Buegler, S. Gamage, R. Atalay, I. Senevirathna, R. Kirste, M. Hoffmann, J. Tweedie, R. Colazzo, A. Hoffmann, Z. Sitar, I. Ferguson, N. Dietz, Wednesday May 25, 11:15am; 38th International Symposium on Compound Semiconductors - ISCS 2011; Berlin GER
21. [Indium-rich InGaN epilayers and heterostructures growth and characterization: the influence of the reactor pressure, growth temperature and surface chemistry on the phase stability,”](#) Nikolaus Dietz, Sampath Gamage, Indika Senevirathna, Ramazan Atalay and Max Buegler, ‘The Workshop on Compound Semiconductor Materials and Devices - WOCSEMMAD 2011,’ February 20-23, 2011, Savannah GA.
22. [The influence of the layer thickness on the optoelectronic properties of InN”](#) Indika Senevirathna, S. Gamage, R. Atalay, J.-I. Hong, N. Dietz, and U. Perera, Thu. Nov. 03, Presentation # EM-ThP-10 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31 - Nov. 04, 2011.
23. [“The effect of reactor pressure on the optoelectronic properties of InN epilayers grown by HPCVD”](#) Indika Senevirathna, S. Gamage, Max Buegler, R. Atalay, J.-I. Hong, N. Dietz, and U. Perera, Presentation # EM-ThP-11 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31 - Nov. 04, 2011.
24. [The influence of ammonia - MO precursors pulse separation on the gallium incorporation in indium-rich In<sub>x</sub>Ga<sub>1-x</sub>N epilayers”](#) Sampath Gamage, R. Atalay, I. Senevirathna, J. Tweedie, R. Collazo and N. Dietz, Presentation # EM-ThP-13 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31, 2011.
25. [“Thermal stability of HPCVD grown InN epilayers”](#) Ananta Acharya, S. Gamage, N. Dietz and B. Thoms, Presentation # EM-ThP-14 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31 - Nov. 04, 2011.
26. [“Substrate template and V/III-ratio effects on the surface and structural properties of HPCVD grown InN films”](#) A. Acharya, M. Buegler; R. Atalay; S. Gamage; J. Tweedie, R. Collazo, N. Dietz and B. Thoms, Presentation # EM-ThP-15 (Electronic Materials and Processing) at AVS 58th International Symposium, Nashville TN, Oct. 31 - Nov. 04, 2011.
27. [“Digital precursor injection approach for improved indium-rich InGa<sub>N</sub> layers grown by HPCVD,”](#) N. Dietz, R. Atalay, M. Buegler, S. Gamage, R. Kirste, M.K.I. Senevirathna, I. Ferguson, J. Tweedie, R. Collazo, Z. Sitar, and A. Hoffmann; Poster# PB2.04, Tuesday, July 12, 2011, 9th International Conference on Nitride Semiconductors, Glasgow UK (2011).
28. [“Optical and structural properties of In<sub>1-x</sub>Ga<sub>x</sub>N epilayers,”](#) R. Atalay, M. Buegler, S. Gamage, I. Senevirathna, J. Wang, and N. Dietz, Contrib. 7784-32, Aug. 4, 2010 - 2:30 pm, at 10th Intern. Conf. on SSL, SPIE San Diego, Aug. 1-5, 2010.
29. [“Reactor pressure - growth temperature relation for InN epilayer grown by high-pressure CVD,”](#) M. Buegler, S. Gamage, ... and N. Dietz, Contrib. 7784-13, Aug. 3, 2010 - 2:20 pm, at 10th Intern. Conf. on SSL, SPIE San Diego, Aug. 1-5, 2010.
30. [“On the pursuit of a common growth window for embedded indium-rich group III-nitride heterostructures,”](#) contributed presentation: N. Dietz, M. Buegler, S. Gamage, ... Contrib. 7784-02, Aug. 2, 2010 at 10th Intern. Conf. on SSL, SPIE San Diego, Aug. 2-5, 2010.

31. “Studies on single-phase, indium-rich In<sub>1-x</sub>Ga<sub>x</sub>N epilayers grown by high-pressure CVD,” contributed presentation: M. Buegler, M. Alevli, R. Atalay, ... I. Ferguson, N. Dietz, APS March Meeting Vol. 55(2), Portland, OR, March 18 (2010).
32. “The Relationship between Surface Termination and Crystal Structure for HPCVD-grown InN Layers,” A.R. Acharya, M. Buegler, R. Collazo, N. Dietz, B. Thoms, Paper EM-TuP2, AVS 57th Intern. Symp. Albuquerque, NM, Oct. 17-22, 2010.
33. “Free Carrier Concentration Analysis in InN and indium-rich In<sub>1-x</sub>Ga<sub>x</sub>N epilayers,” R. Kirste, S. Mohn, M. Buegler, ... N. Dietz, A. Hoffmann, Int. Workshop on Nitride Semiconductors (IWNS2010), Tampa FL Sept. 19-24, 2010, Contrib. Poster 904278, Sept. 22, 2010.
34. “The Influence of the Reactor Pressure on the Growth Temperature of InN epilayers and their physical properties,” S. Gamage, ... N. Dietz, Int. Workshop on Nitride Semiconductors (IWNS2010), Tampa FL Sept. 19-24, 2010, Contrib. Poster 904314, Sept. 20, 2010.
35. “Influence of Reactor Pressure on the Phase Stability of indium-rich In<sub>1-x</sub>Ga<sub>x</sub>N epilayers,” M. Buegler, ... I. Ferguson, R. Collazo, Z. Sitar, and N. Dietz - at 15th Int. Conf. on MOVPE, Hyatt Regency, Lake Tahoe, May 27, 2010.

#### ***Conference - poster contributions***

1. “Effect of Photoluminescent Indium Nitride Nanocrystals on the Performance of a-Si:H Solar Cell,” F. I. Chowdhury, K. Islam, S. Alkis, V. Kumar, B. Ortac, M. Alevli, N. Dietz, A. K. Okay, A. Nayfeh; MSR Fall Meeting 2014 Symposium II: Semiconductor Nanocrystals, Plasmonic Metal Nanoparticles, and Metal-Hybrid Structures, poster IIS.32, Dec 2, 2014 (2014)
2. “Indium Nitride Nanocrystals Obtained through Laser Ablation for Large Area Optoelectronics,” B. Tekcan, S. Alkis, M. Alevli, N. Dietz, B. Ortac, N. Biyikli, A. K. Okay; MSR Fall 2014 Symposium AA9: Poster Session: Synthesis, Plasticity, and Theory, poster AA9.02, Dec. 3, 2014, Symposium AA: Synthesis, Processing and Mechanical Properties of Functional Hexagonal Materials (2014)
3. “Effects of Substrate Polarity on the Physical Properties of InN Epilayers Grown at Super-Atmospheric Pressures,” S. Gamage, R. Kirste, M.K.I. Senevirathna, F. Kaess, M. Bobea, R. Collazo, Z. Sitar, and N. Dietz; MSR Fall Meeting 2014 Symposium AA9: Poster Session: Synthesis, Plasticity, and Theory, poster AA9.13, Dec. 3, 2014, Symposium AA: Synthesis, Processing and Mechanical Properties of Functional Hexagonal Materials (2014)
4. “In Situ Metrology during GaN and InGa<sub>N</sub> Growth by Remote Plasma-assisted MOCVD,” D. Seidlitz, R. Samaraweera, I.T. Ferguson, N. Dietz and A. Hoffmann, Paper# EM-TuP15, Tue. Nov. 11, 2014, session “Electronic Materials and Processing Poster Session,” AVS 61st International Symposium; Baltimore, MD, Nov. 9-11, 2014.
5. “Influence of Plasma-Activated Nitrogen Species in MOCVD Grown GaN/GaInN Epilayers,” R. Samaraweera, D. Seidlitz, M.K.I. Senevirathna, B. Hussain, I.T. Ferguson, and N. Dietz; Paper# EM-TuP17, Tue. Nov. 11, session “Electronic Materials and Processing Poster Session,” AVS 61st International Symposium; Baltimore, MD, Nov. 9-14 2014.

6. “Properties of InN epilayers grown at superatmospheric reactor pressures,” S. Gamage, I.M. Senevirathna, B. Kucukgok, B. Hussain, I. Ferguson, and N. Dietz, Symp. SPIE Opt. Eng. & Appl., Aug. 20, 2014, SPIE Paper# 9190-41, 13<sup>th</sup> Int. Conf. on SSL and LED-based Illumin. Systems; San Diego, CA (2014).
7. “Structural and optical properties of InN epilayers grown at superatmospheric reactor pressures,” S. Gamage, M.K.I. Senevirathna, B. Kucukgok, B. Hussain, I. Ferguson, and N. Dietz; 5th Int. Conf. White LEDs and Solid State Lighting (WLED-5), Paper# A1029 (Poster), June 1-5, Samdo2-dong, Jeju City, Jeju-do, Korea (2014); <http://www.wled5.org/>
8. “Optoelectronic applications of ultrasmall size InN nanoparticles obtained by laser ablation of high pressure chemical vapor deposition (HPCVD) grown InN thin Film,” Sabri Alkis, Burak Tekcan, Mustafa Alevli, Nikolaus Dietz, Ali Kemal Okyay, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), Poster presentation, paper# E24, May 19, 6-8pm, Atlanta, GA (2014).
9. “Real-time optical growth characterization of group III-nitride-alloys during Plasma-Assisted MOCVD,” D. Seidlitz, R. Samaraweera, B. Hussain, I. Ferguson, N. Dietz and A. Hoffmann, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), Poster presentation, paper# J27 - contr# 1966966, May 20, 6-8pm, Atlanta, GA (2014).
10. “Growth Temperature and Free Carrier Correlations in InN Studied by FTIR and Photoluminescence,” M.K.I. Senevirathna, S. Gamage, R. Samaraweera, M. Bugler, A. Hoffmann, A.G.U. Perera, and N. Dietz, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), Poster presentation, paper# E26 - contr# 1968162, May 19, 6-8pm, Atlanta, GA (2014).
11. “Plasma-Assisted MOCVD growth of GaN and InGaN epilayers,” R. Samaraweera, D. Seidlitz, B. Hussain, A. Melton, I. Senevirathna, I. Ferguson, and N. Dietz, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), Poster presentation, paper# E8, Contr# 1967981, May 19, 6-8pm, Atlanta, GA (2014).
12. “Structural, Optical and electrical properties of InN epilayers grown at super-atmospheric pressures,” S. Gamage, M.K.I. Senevirathna, M. Büegler, M.A.R.L. Samaraweera, K. Nanayakkara, A. Hoffman and N. Dietz, 5th Int. Symp. Growth of III-Nitrides (ISGN-5), Poster presentation, paper# E7, Contr# 1968006, May 19, 6-8pm, Atlanta, GA (2014).
13. “Migration-Enhanced, Remote-Plasma MOVCD Growth of Group III-Nitride Epilayers,” R. Samaraweera, F. Güth, A. Melton, K. Nanayakkara, I. Seneviratne, I. Ferguson and N. Dietz; Aug. 27, 2013; Paper# AP2.23; 10<sup>th</sup> Intern. Conf. on Nitride Semicond. (ICNS10), Aug. 25-30, Washington DC (2013).
14. “Properties of indium-rich InGaN epilayers grown at superatmospheric pressures,” S. Gamage, K. Nanayakkara, I. Senevirathna, A. Melton, I. Ferguson, and N. Dietz; Aug. 26, 2013; Paper# AP1.24; 10<sup>th</sup> Intern. Conf. on Nitride Semicond. (ICNS10), Aug. 25-30, Washington DC (2013).
15. “Structural and Optoelectrical Properties of InN Epilayers Grown by High-Pressure CVD,” A. Acharya, M. Buegler, S. Gamage, N. Dietz, and B. Thoms; Thur. Nov. 01, 2012; contribution#EM-ThP5 (Electronic Materials and Processing) at AVS 59th International Symposium, Tampa, FL, 2012.

16. “[Structural, Compositional, and Thermal Stability Studies on  \$\text{In}\_{1-x}\text{Ga}\_x\text{N}\$  Epilayers](#),” N. Dietz, M. Buegler, S. Gamage, M. K. I. Senevirathna, R. Atalay, B. Kucukgok, A. G. Melton, and I. T. Ferguson; Thur. Nov. 01, 2012; contribution# EM-ThP6 (Electronic Materials and Processing) at AVS 59th International Symposium, Tampa, FL, 2012.
17. “[Effect of the nucleation layer thickness on the physical properties of epitaxial InN layers](#)” S. Gamage, M.K.I. Senevirathna, R. Atalay, A. G. Melton, I.T. Ferguson and N. Dietz; Twelfth International Conference on Solid State Lighting; Conference OP220; August 2012, San Diego, CA, poster# 8484-30, Aug. 13, 5-7:30pm (2012).
18. “[Atomic Layer Epitaxy of InN Films](#),” N. Mahadik, J. K. Hite, M. A. Mastro, C. R. Eddy, Jr., N. Nepal, M. Currie, S. Gamage, I. Senevirathna, N. Dietz, Poster Session A, Tuesday, June 19<sup>th</sup>; 12<sup>th</sup> International Conference on Atomic Layer Deposition (ALD 2012), Dresden Germany, June 17-20, 2012.

1.

**1. Report Type**

Final Report

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**Grant/Contract Title**

The full title of the funded effort.

STABILIZATION OF INDIUM-RICH IN<sub>1-x</sub>GAXN HETEROSTRUCTURES - THE  
EXPLORATION OF A COMMON PROCESSING WINDOW**Grant/Contract Number**

AFOSR assigned control number. It must begin with "FA9550" or "F49620" or "FA2386".

FA9550-10-1-0097

**Principal Investigator Name**

The full name of the principal investigator on the grant or contract.

Nikolaus Detz

**Program Manager**

The AFOSR Program Manager currently assigned to the award

Dr. Kenneth C. Goretta

**Reporting Period Start Date**

04/01/2010

**Reporting Period End Date**

03/31/2015

**Abstract**

During the grant period, the growth and optimization of indium-rich In<sub>1-x</sub>GaxN layers grown by high-pressure metal organic chemical vapor deposition (MOCVD) was explored at reactor pressures from 5 to 20 bar and at growth temperatures of 700-900 °C. The goal was to evaluate the reactor pressure and growth temperature relation at which indium-rich In<sub>1-x</sub>GaxN layers can be stabilized. The results showed that for pressures around 15 bar, the growth temperatures for InGaN varies from 850 °C (InN) to 950 °C (In<sub>0.7</sub>Ga<sub>0.3</sub>N), significantly reducing the temperature gap in the ternary InGaN system compared to low-pressure MOCVD. An unexpected side effect found was the significant reduction in growth rate with increasing reactor pressures, which is due to smaller surface diffusion layers at higher pressures. The results on forming single phase indium-rich ternary InGaN alloys using simultaneous and sequential group-III precursor injection sequences was only partially successful: We obtained single phase alloys for In<sub>1-x</sub>GaxN [0 < x < 0.15] and [0.25 < x < 0.3] but observed mixed phases for compositions between for the designating concept explored. The experiments indicated the presence of Ga- and/or In-



ad layers - and oscillations between them – that may play a major role for the observed mixed InGaN phases. Additional studies will be needed to obtain a better understanding on how the deposition of precursors to the surface relates to the surface decomposition and chemistry processes that influence the Ga- and In-fragment incorporation and the subsequent InGaN phase formation.

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### Archival Publications (published) during reporting period:

2. "A Near-Infrared Range Photodetector Based on Indium Nitride Nanocrystals Obtained Through Laser Ablation," B. Tekcan, S. Akş, M. Aev, N. Detz, B. Ortaç, N. B y k, and A.K. Okyay, IEEE Electron Device Lett. 35 (9) pp.936-9, Sept. 2014; doi : 10.1109/LED.2014.2336795
3. "Enhanced memory effect via quantum confinement in 16 nm InN nanoparticles embedded in ZnO charge trapping layer," N. E -Atab, F. C men, S. Akş, B. Ortaç, M. Aev, N. Detz, A.K. Okyay, and A. Nayfeh, Appl. Phys. Lett. 104(25), p.253106 (2014); doi .org/10.1063/1.4885397
4. "The Group III-Nitride Material Class: from Preparation to Perspectives in Photoelectrocatalysis," R. Coz and N. Detz, book chapter 8 in 'Photoelectrochemical Water Splitting: Issues and Perspectives,' ed. H-J. Lewerenz and L.M. Peter, RSC Publishing, pp. 193-222, (2013). <http://dx.doi.org/10.1039/9781849737739-00193>
5. "Thermal stability of InN epilayers grown by high pressure chemical vapor deposition," A. Acharya, S. Gamage, M. Senevirathne, M. Aev, K. Bahadur, A. Me ton, I. Ferguson, N. Detz, and B. Thoms; Appl. Surf. Sci. 268, p.1 (2013). doi .org/10.1016/j.apsusc.2012.10.184
6. "Room Temperature GaN-based Spin Polarized Emitter," A. G. Me ton, B. Kucukgok, Z. Lu, N. Detz, N. Lu and I. T. Ferguson, Proc. SPIE Vol. 8631, pp.863104-1-9 (2013). doi : 10.1117/12.2012586
7. "Development of indium-rich InGaN epilayers for integrated tandem solar cells," A. G. Me ton, B. Kucukgok, B-Z. Wang, N. Detz, N. Lu and I. T. Ferguson, Mater. Res. Soc. Symp. Proc. Vol. 1493, E15.02, pp.1-6 (2013). doi :10.1557/op .2013.229
8. "Room Temperature GaN-based Spin Polarized Emitter," A. G. Me ton, B. Kucukgok, Z. Lu, N. Detz, N. Lu and I. T. Ferguson, Proc. SPIE Vol. 8631, pp.863104-1-9 (2013). doi : 10.1117/12.2012586
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10. Thermal stability of InN epilayers grown by high pressure chemical vapor deposition," A. Acharya, S. Gamage, M. Senevirathne, M. Aev, K. Bahadur, A. Me ton, I. Ferguson, N.

- D etz, B. Thoms, App . Surf. Sc . 268, p.1 (2013). do .org/10.1016/j.apsusc.2012.10.184
11. "Effect of nuc eat on per od on the phys ca propert es of InN ep ayers," S. Gamage, M. K. I. Senev rathna, R. Ata ay, A. G. U. Perera, A. G. Me ton, I. T. Ferguson and N. D etz, Proc. of SPIE Vo . 8484 pp.84841I-5 (2012). do .org/10.1117/12.930363
  12. "Effect of V/III mo ar rat o on the structura and opt ca propert es of InN ep ayers grown by HPCVD," R. Ata ay, M. Bueg er, S. Gamage, M. Senev rathna, B. Küçükgök, A. Me ton, A. Hoffmann, A. Perera, I. Ferguson and N. D etz, Proc. of SPIE Vo . 8484 pp.84840X-8 (2012). do .org/10.1117/12.930199
  13. "Effect of reactor pressure on the e ectr ca and structura propert es of InN ep ayers grown by h gh-pressure chem ca vapor depos t on," M. Senev rathna, S. Gamage, R. Ata ay, A. R. Acharya, A. Perera, N. D etz, M. Bueg er, A. Hoffmann, L. Su, A. Me ton, and I. Ferguson, J. Vac. Sc . Techno . A 30(3), pp.031511-6 (2012). do :10.1116/1.4705727
  14. "Observat on of NH<sub>2</sub> spec es on t ted InN(01-11) facets," A. R. Acharya, M. Bueg er, R. Ata ay, N. D etz, B. D. Thoms, J.S. Tweed e and R. Co azo, J. Vac. Sc . Techno . A 29(4) pp.041402-5 (2011).
  15. "Growth temperature and growth rate dependency on reactor pressure for InN ep ayers grown by HPCVD," M. Bueg er, S. Gamage, R. Ata ay, J. Wang, M. K. I. Senev rathna, R. K rste, T. Xu, M. Jam , I. Ferguson, J. Tweed e, R. Co azo, A. Hoffmann, Z. S tar, and N. D etz, Phys. Stat. So . (c) 8 pp. 2059-2062 (2011).
  16. "Reactor pressure - growth temperature re at on for InN ep ayers grown by h gh-pressure CVD," M. Bueg er, S. Gamage, R. Ata ay, J. Wang, I. Senev rathna, I. Ferguson, R. Co azo, A. Hoffmann, Z. S tar, and N. D etz, Proc. of SPIE Vo . 7784, do : 10.1117/12.860952, 77840F-1-7 (2010).
  17. "Opt ca Propert es of InN Grown on Temp ates w th Contro ed Surface Po ar t es," R. K rste, M.R. Wagner, A. Str ttmatter, J. H. Schu ze, R. Co azo, S. S tar , M. A ev , N. D etz and A. Hoffmann, phys ca status so d (a), 207(10) pp. 2351–2354 (2010).
  18. "The effects of V/III mo ar rat o on structura propert es of In<sub>0.65</sub>Ga<sub>0.35</sub>N ayers grown by HPCVD," G. Durkaya, M. Bueg er, R. Ata ay, I. Senev rathne, M. A ev , O. H tzemann, M. Ka ser, R. K rste, A. Hoffmann, and N. D etz, phys ca status so d (a) 207, pp. 1379 (2010).
  19. "Growth temperature - phase stab ty re at on n In<sub>1-x</sub>GaxN ep ayers grown by h gh-pressure CVD," G. Durkaya, M. A ev , M. Bueg er, R. Ata ay, S. Gamage, M. Ka ser, R. K rste, A. Hoffmann, M. Jam , I. Ferguson, N. D etz, Mater. Res. Soc. Symp. Proc. 1202, paper# 1202-I05-21, pp.1-6 (2010).

**Changes in research objectives (if any):**

None

**Change in AFOSR Program Manager, if any:**

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**Extensions granted or milestones slipped, if any:**

None

**AFOSR LRIR Number**

**LRIR Title**

**Reporting Period**

**Laboratory Task Manager**

**Program Officer**



## Research Objectives

## Technical Summary

## Funding Summary by Cost Category (by FY, \$K)

	Start ng FY	FY+1	FY+2
Salary			
Equipment/Facilities			
Supplies			
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